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Journal of Solid State Chemistry 175 (2003) 316–321

JOURNAL OF  
SOLID STATE  
CHEMISTRY

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# A reinvestigation of quaternary layered bismuth oxyhalides of the Sillén X1 type

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Received 3 March 2003; received in revised form 15 May 2003; accepted 17 May 2003

## Abstract

A family of layered bismuth oxyhalides,  $L_{0.5}^{I}Bi_{1.5}O_2X$  and  $L^{II}BiO_2X$  has been reinvestigated. Formation of X1-type Sillén compounds has been established for  $L^I = \text{Li, Na}$ ,  $L^{II} = \text{Ca, Sr, Ba}$ , and  $X = \text{Cl, Br, I}$ , but the details of their crystal structures are different. While all  $L_{0.5}^{I}Bi_{1.5}O_2X$ ,  $\text{CaBiO}_2\text{Br}$ , and  $\text{CaBiO}_2\text{I}$  adopt the disordered tetragonal  $\text{Nd}_2\text{O}_2\text{Te}$  structure, all compounds of  $L^{II} = \text{Sr}$  and  $\text{Ba}$  are orthorhombic and isostructural to  $\text{PbSbO}_2\text{Cl}$ , due to  $L/\text{Bi}$  cation ordering. Crystal structures have been determined for  $\text{CaBiO}_2\text{I}$ ,  $\text{SrBiO}_2\text{Br}$ ,  $\text{SrBiO}_2\text{I}$ , and  $\text{BaBiO}_2\text{I}$ . We discuss the factors which determine the occurrence and type of cation ordering in the quaternary bismuth and antimony X1-type oxyhalides. We also predict that more isostructural compounds can be prepared with antimony.

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**Keywords:** Bismuth oxyhalides; Crystal structures; Cation ordering; Sillén phases

## 1. Introduction

The first members of the Sillén layered family of bismuth oxyhalides were reported as early as 1939 [1,2]; the family now comprises oxyhalides of Bi with Li, Na, Ca, Sr, Ba, Pb, and Cd [3]. Their crystal structures can be regarded as ordered alternation of fluorite-derived [ $M_2O_2$ ] slices (where  $M = \text{Bi}$  and its substituents listed above) and single or double halide ion sheets. These layer sequences give rise to ideal tetragonal structures with  $a \sim 3.9 \text{ \AA}$  (i.e.,  $\sqrt{2/2a}$ (fluorite)) and  $c$  varying from 6 to 50 Å dependent on structural complexity. Despite numerous investigations in this area (dedicated both to their crystal chemistry [3–10] and catalytic activity in oxidative coupling of methane [11]) many structural details are as yet unclear. The compounds with the simplest “X1” structure, which corresponds to alternation of [ $M_2O_2$ ]<sup>+</sup> slices and [ $X$ ]<sup>−</sup> halide ion sheets, were initially described as tetragonal, i.e., adopting the ideal structure (which is an anti-type to  $\text{ThCr}_2\text{Si}_2$  [12]). However, as yet only  $\text{Na}_{0.5}\text{Bi}_{1.5}\text{O}_2\text{Cl}$  has been reported

recently as truly tetragonal (Fig. 1a) [13]. This structure was also reported for  $\text{Bi}_2\text{O}_2\text{Se}$  [14] and  $\text{Nd}_2\text{O}_2\text{Te}$  [15]. The structures of the oxychlorides with divalent elements,  $L^{II}\text{BiO}_2\text{Cl}$  ( $L^{II} = \text{Cd, Ca, Sr, Ba}$ ) demonstrate two different versions of complete ordering of the  $L^{II}$  and  $\text{Bi}^{III}$  cations contributing to the fluorite layers. The smallest  $\text{Ca}^{2+}$  [9] and  $\text{Cd}^{2+}$  [10] give a monoclinic structure (Fig. 1b) while the larger  $\text{Sr}^{2+}$  [9] and  $\text{Ba}^{2+}$  [8] result in an orthorhombic superstructure (Fig. 1c) found earlier for the related  $\text{BaSbO}_2\text{Cl}$  [16],  $\text{PbSbO}_2\text{Cl}$  [7,17,18], and  $\text{PbBiO}_2\text{Cl}$  [4]. The lead compounds are peculiar.  $\text{PbBiO}_2\text{Cl}$  crystallizes in the orthorhombic structure as the natural mineral perite [4]; however, synthetic perite is always tetragonal [6].  $\text{PbSbO}_2\text{Cl}$  may be converted from orthorhombic to tetragonal form by treatment with boiling water [17]; more recent investigation has shown that stoichiometric  $\text{PbSbO}_2\text{Cl}$  and  $\text{PbSbO}_2\text{Br}$  are orthorhombic but a shift of composition to  $\text{PbSbO}_{2.125}\text{X}_{0.75}$  makes the compounds tetragonal [7]. Stoichiometric  $\text{PbSbO}_2\text{I}$  is also tetragonal, akin to  $\text{PbBiO}_2\text{Br}$  and  $\text{PbBiO}_2\text{I}$  [7]. Though both stoichiometry and size of the contributing ions are likely to affect the structural details of these X1-type Bi and Sb oxyhalides, neither the oxybromides nor oxyiodides of bismuth and

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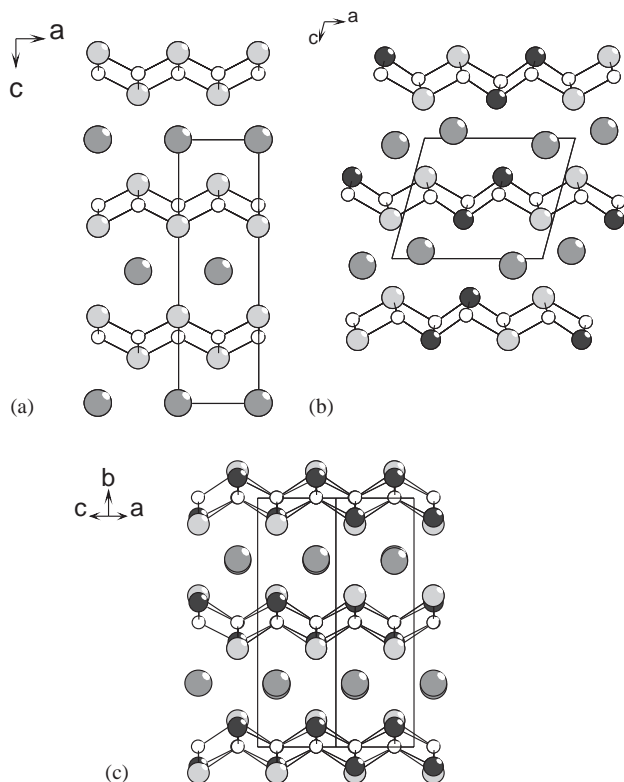


Fig. 1. Crystal structures of the three variants of the Sillen X1 structure type. (a) Tetragonal form ( $\text{Nd}_2\text{O}_2\text{Te}$ -type), (b) monoclinic form ( $\text{CaBiO}_2\text{Cl}$ -type), (c) orthorhombic form ( $\text{PbSbO}_2\text{Cl}$ -type). Note the differing cation ordering schemes for (b) and (c).

alkali and alkaline earth metals, to the best of our knowledge, have been studied in detail since 1941. Moreover, we could not find any reference to existence of  $\text{CaBiO}_2X$  ( $X = \text{Br}$  and  $\text{I}$ ) and  $\text{SrBiO}_2\text{I}$ ;  $\text{BaBiO}_2\text{I}$  was mentioned briefly in Ref. [19], but no data were presented for this compound.

Upon preparation of  $\text{BaBiO}_2\text{Br}$  as a precursor for more complex bismuth oxyhalides [20], we have noticed that it is orthorhombic, similar to  $\text{BaBiO}_2\text{Cl}$  [7] and not tetragonal as reported originally by Sillén [2]. A question arose whether other X1 compounds described originally as tetragonal may actually have lower symmetry. Representatives of this family might also exist among oxyhalides of bismuth and some other  $L$  cations. Hence, the whole series of  $L_{0.5}\text{Bi}_{1.5}\text{O}_2X$  and  $L^{\text{II}}\text{BiO}_2X$  oxyhalides ( $X = \text{Br}$ ,  $\text{I}$ ) has been reinvestigated, excluding  $L^{\text{II}}\text{BiO}_2\text{Cl}$  and Pb compounds (which have been studied recently [6–10]). Preliminary results of our investigation have been reported in Ref. [21]. Besides the known  $L = \text{Li}$ ,  $\text{Na}$ ,  $\text{Ca}$ ,  $\text{Cd}$ ,  $\text{Sr}$ , and  $\text{Ba}$ , we also considered  $L = \text{Sn}^{\text{II}}$ ,  $\text{K}$ ,  $\text{Cu}^{\text{I}}$ , and  $\text{Ag}$ .

<sup>1</sup>There is a misprint in Ref. [3]: the cell parameters attributed to  $\text{CaBiO}_2\text{I}$  actually belong to  $\text{CdBiO}_2\text{I}$  [2].

## 2. Experimental

$\text{CaBiO}_2X$  and  $\text{SrBiO}_2X$  ( $X = \text{Br}$ ,  $\text{I}$ ) were prepared directly from  $LO$  and  $\text{BiOX}$  in alumina crucibles sealed into evacuated silica ampoules by annealing the corresponding 1:1 mixtures at  $750^\circ\text{C}$  ( $\text{Ca}$ ) or  $800^\circ\text{C}$  ( $\text{Sr}$ ) for 5 days.  $\text{CaO}$  and  $\text{SrO}$  were obtained by calcining the carbonates at  $1100^\circ\text{C}$  for 2 days. The barium compounds were prepared in a two-step process. First, a 1:1 mixture of  $\text{BaCO}_3$  and  $\text{Bi}_2\text{O}_3$  was annealed in dynamic vacuum at  $\sim 800^\circ\text{C}$  for 8 h. The resulting dark red precursor was mixed with anhydrous  $\text{BaX}_2$  (prepared from  $\text{BaX}_2 \cdot 2\text{H}_2\text{O}$  at  $250^\circ\text{C}$  in dynamic vacuum) and annealed under the same conditions as for  $\text{SrBiO}_2X$ .

In the search for  $\text{SnBiO}_2X$  ( $X = \text{Cl}$ ,  $\text{Br}$ ,  $\text{I}$ ),  $\text{SnO}$  and  $\text{BiOX}$  (1:1) were annealed in evacuated silica tubes at  $700^\circ\text{C}$  for 7 days. The precursors were prepared according to [22]. For synthesis of  $\text{CdBiO}_2X$  ( $X = \text{Br}$ ,  $\text{I}$ ),  $\text{CdO}$  and  $\text{BiOX}$  were annealed at the same conditions as given for  $\text{CdBiO}_2\text{Cl}$  [10].

$\text{Na}_{0.5}\text{Bi}_{1.5}\text{O}_2X$  were obtained directly from 1:1:1 mixtures of  $\text{NaX}$  ( $\text{NaI} \cdot 2\text{H}_2\text{O}$  was dehydrated under the same conditions as  $\text{BaX}_2$ ),  $\text{BiOX}$  and  $\text{Bi}_2\text{O}_3$  in evacuated silica tubes at  $650^\circ\text{C}$  for 10 days. The potassium, copper and silver-containing samples were treated the same way, but for the  $d$ -metals, the annealing temperature was lower ( $400$ – $450^\circ\text{C}$ ). The lithium compounds were synthesized from  $\text{LiBiO}_2$  and  $\text{BiOX}$  in alumina crucibles sealed into silica ampoules and annealed at  $650^\circ\text{C}$ .  $\text{LiBiO}_2$  in its turn was prepared by melting  $\text{Li}_2\text{CO}_3$  and  $\text{Bi}_2\text{O}_3$  in an alumina crucible under dynamic vacuum for 30 min.

The phase purity of all samples was monitored by X-ray powder diffraction using a Guinier camera (FR-552, Enraf-Nonius,  $\text{CuK}\alpha_1$  radiation). The reflection positions on the Guinier films were determined with an accuracy of  $0.005\text{ mm}$ , converted into interplanar spacings, and cell parameters of the obtained compounds were refined using the whole pattern by least-squares fits using a home-made program.

## 3. Results

The X-ray diffraction data indicated formation of X1-type compounds only for  $L^{\text{I}} = \text{Li}$  and  $\text{Na}$  and  $L^{\text{II}} = \text{Ca}$ ,  $\text{Sr}$ , and  $\text{Ba}$  with all three halogens. The compounds of  $\text{Li}$ ,  $\text{Na}$  were found to be tetragonal, in fair agreement with Sillén's original reports [1,2];  $\text{CaBiO}_2\text{Br}$  and  $\text{CaBiO}_2\text{I}$  are also tetragonal. On the contrary, all compounds of  $\text{Sr}$  and  $\text{Ba}$  appeared to be orthorhombic ( $\text{PbSbO}_2\text{Cl}$  type). No new example of the monoclinic  $\text{CaBiO}_2\text{Cl}$  structure type [9] has been found. The cell parameters of the compounds which were obtained single phase are listed in Table 1. A small amount of  $\text{BiOI}$  was found in the  $\text{Li}_{0.5}\text{Bi}_{1.5}\text{O}_2\text{I}$  sample;

Table 1  
Lattice parameters for the X1 bismuth oxyhalides

(a) Structure type Nd <sub>2</sub> O <sub>2</sub> Te—SG <i>I4/mmm</i> , <i>Z</i> = 2					
Compound	<i>a</i> (Å)	<i>c</i> (Å)	Compound	<i>a</i> (Å)	<i>c</i> (Å)
Li <sub>0.5</sub> Bi <sub>1.5</sub> O <sub>2</sub> Cl	3.8457(5)	12.060(3)	Na <sub>0.5</sub> Bi <sub>1.5</sub> O <sub>2</sub> Br	3.9205(3)	12.542(2)
Li <sub>0.5</sub> Bi <sub>1.5</sub> O <sub>2</sub> Br	3.8789(5)	12.486(2)	Na <sub>0.5</sub> Bi <sub>1.5</sub> O <sub>2</sub> I	3.9987(5)	13.322(3)
Li <sub>0.5</sub> Bi <sub>1.5</sub> O <sub>2</sub> I <sup>a</sup>	3.935(2)	13.236(4)	CaBiO <sub>2</sub> Br	3.9617(5)	12.584(3)
Na <sub>0.5</sub> Bi <sub>1.5</sub> O <sub>2</sub> Cl	3.880(1)	12.136(4)	CaBiO <sub>2</sub> I	4.0222(2)	13.239(1)
(b) Structure type PbSbO <sub>2</sub> Cl—SG <i>Cmcm</i> , <i>Z</i> = 4					
Compound	<i>a</i> (Å)	<i>b</i> (Å)	<i>c</i> (Å)		
SrBiO <sub>2</sub> Br	5.768(2)	12.806(4)	5.645(2)		
SrBiO <sub>2</sub> I	5.776(3)	13.509(8)	5.863(2)		
BaBiO <sub>2</sub> Br	5.968(2)	13.162(6)	5.748(2)		
BaBiO <sub>2</sub> I	6.052(2)	14.031(6)	5.825(3)		

<sup>a</sup> BiOI is present as an impurity phase.

the composition of the target phase was therefore attributed by analogy with the isostructural compounds.

Despite numerous efforts, CdBiO<sub>2</sub>Br and CdBiO<sub>2</sub>I could not be prepared single-phase; admixtures of X3-type phases (which seem to be more thermally stable compared to X1) were always present. However, powder patterns of X1-richest samples indicated truly tetragonal symmetry akin to the corresponding calcium compounds (no superstructure reflections or splitting of the strongest peaks characteristic for ordered structures have been detected).

The samples aimed at SnBiO<sub>2</sub>X appeared black, and X-ray diffraction data indicated presence of SnO<sub>2</sub>, Bi<sub>2</sub>Sn<sub>2</sub>O<sub>7</sub> and Bi which means a redox reaction had actually proceeded. The search for K<sub>0.5</sub>Bi<sub>1.5</sub>O<sub>2</sub>X yielded mixtures of KX and solid solutions based on Bi<sub>24</sub>O<sub>31</sub>X<sub>10</sub> (*X* = Cl, Br) or Bi<sub>7</sub>O<sub>9</sub>I<sub>3</sub>. Diffraction patterns of samples with Cu and Ag yielded mixtures of unknown phases with *X* = Cl and I. For *X* = Br, the patterns appeared similar to each other and could be, with moderate success, indexed on the basis of I-centered tetragonal cells with *a* ~ 3.9 Å and *c* ~ 22 Å indicating the existence of a novel, possibly layered bismuth oxyhalide family. Our attempts to prepare tetragonal L<sup>II</sup>BiO<sub>2.125</sub>X<sub>0.75</sub> with L<sup>II</sup> = Sr and Ba by analogy with PbSbO<sub>2.125</sub>X<sub>0.75</sub> also proved unsuccessful, resulting in mixtures of orthorhombic L<sup>II</sup>BiO<sub>2</sub>X and unknown phases.

#### 4. Crystal structures

Structural details have been determined for tetragonal CaBiO<sub>2</sub>I and orthorhombic SrBiO<sub>2</sub>Br, SrBiO<sub>2</sub>I, and BaBiO<sub>2</sub>I. The experimental conditions are gathered in Table 2. The powder data sets were collected on a Stoe STADI/P diffractometer using monochromatized CuKα<sub>1</sub> radiation. Some samples were attached to the holder with vaseline oil; in such cases, two small 2θ regions containing the strongest oil reflections have been

excluded from the refinement. Atomic coordinates of tetragonal PbBiO<sub>2</sub>I [6] and orthorhombic BaBiO<sub>2</sub>Cl [8] served as initial parameters. The GSAS package software [23] was used for all calculations. Two typical Rietveld refinement plots, one for tetragonal CaBiO<sub>2</sub>I and the other for orthorhombic SrBiO<sub>2</sub>I, are presented in Figs. 2a and b. The atomic coordinates and selected bond distances are given in Tables 3 and 4, respectively.

There are just minor differences in bond distances between isotopic CaBiO<sub>2</sub>I and PbBiO<sub>2</sub>I [6]. The (L,Bi)–O bond is 0.08 Å shorter but (L,Bi)–I distance 0.02 Å longer in CaBiO<sub>2</sub>I compared to PbBiO<sub>2</sub>I. It is curious to note that in the orthorhombic SrBiO<sub>2</sub>Cl–SrBiO<sub>2</sub>Br–SrBiO<sub>2</sub>I series, the Sr–*X*, Bi–*X*, and Bi–O distances elongate as expected but not Sr–O which shortens slightly when passing from *X* = Cl to Br and I. This shortening is even more pronounced comparing the structures of BaBiO<sub>2</sub>Cl and BaBiO<sub>2</sub>I (0.11 Å). When passing from SrBiO<sub>2</sub>I to BaBiO<sub>2</sub>I, both Bi–O and Bi–I distances elongate (by ~0.04 and ~0.15 Å, respectively). The latter value, together with the Bi–I distances themselves indicates very little or no bonding between Bi and I, at least, in BaBiO<sub>2</sub>I.

#### 5. Discussion

Our investigations have added several new representatives to the X1 bismuth oxyhalide family. Besides *L* = Pb, these structures form only with *L* = Li, Na, Ca, Cd, Sr and Ba, in fair agreement with Sillén's results [2,24]. These cations readily adopt the same coordination environment as Bi<sup>3+</sup>, and difference between their radii and that of Bi<sup>3+</sup> does not exceed 0.25 Å [25]. If the difference in ionic radii is larger or the coordination preferences are different, the X1 structure is unlikely to occur, which is the case for *L* = K, Cu, or Ag. Sn<sup>2+</sup> meets both these requirements but is perhaps a too

Table 2  
Experimental conditions for the powder diffraction experiments

Compound	CaBiO <sub>2</sub> I	SrBiO <sub>2</sub> Br	SrBiO <sub>2</sub> I	BaBiO <sub>2</sub> I
Formula weight	408.0	408.5	455.5	505.1
Crystal system	Tetragonal		Orthorhombic	
Space group	<i>I4/mmm</i> (# 139)		<i>Cmcm</i> (# 63)	
Cell parameters:				
<i>a</i> (Å)	4.01943(4)	5.7676(1)	5.8723(1)	6.0492(1)
<i>b</i> (Å)	4.01943(4)	12.8246(4)	13.5344(2)	14.0224(2)
<i>c</i> (Å)	13.2144(1)	5.6460(1)	5.7363(1)	5.8212(1)
<i>V</i> (Å <sup>3</sup> )	213.490(4)	417.62(2)	455.91(2)	493.78(1)
<i>Z</i>	2		4	
Calculated density	6.346	6.497	6.636	6.796
Diffractometer		STOE STADI/P		
Radiation	CuKα <sub>1</sub> (λ = 1.54059 Å)			
2θ range (deg)	5–100	12–120	5–100	10–100
Number of reflections	48	198	151	199
Number of data points	9499	5338 <sup>a</sup>	9499	9919
Number of free parameters	18	27	23	25
Analyzing package		GSAS		
<i>R</i> ( <i>F</i> <sup>2</sup> )	0.0952	0.0748	0.0858	0.0930
<i>R</i> <sub>p</sub>	0.1011	0.0880	0.0994	0.0545
<i>R</i> <sub>wp</sub>	0.1376	0.1200	0.1349	0.0770
χ <sup>2</sup>	2.29	2.61	2.26	6.75

<sup>a</sup> The 2θ step size is 0.02°, for others 0.01°.

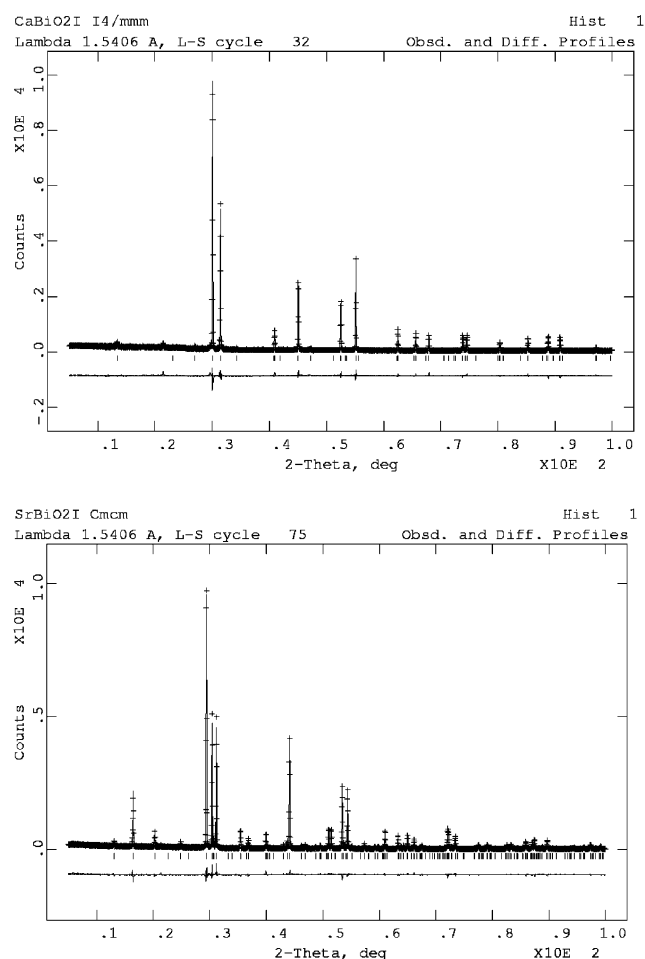


Fig. 2. Final Rietveld plots for (a) CaBiO<sub>2</sub>I, (b) SrBiO<sub>2</sub>I.

strong reducing agent to neighbor Bi<sup>3+</sup> in the same structure.

The occurrence and type of cation ordering in the X1-type bismuth (and perhaps antimony) oxyhalides depends on at least three parameters: the overall stoichiometry of the fluorite layers, the relative size of the ions contributing to the structure, and their electronic structure. The ordering occurs only in the case of a *L*:Bi (or *L*:Sb) ratio of 1:1. As soon as the interlayer charge balance requires non-integer stoichiometry of the fluorite layers (0.5:1.5 in case of *L*<sup>1</sup>), only the ideal Nd<sub>2</sub>O<sub>2</sub>Te structure is realized. This is consistent with structural data for bismuth oxyhalides with related structures, where Bi<sup>3+</sup> in the [Bi<sub>2-*x*L<sub>x</sub>O<sub>2</sub>]<sub>2</sub> layers is substituted by lower-valent *L* cations so that *x* is non-integer: the overwhelming majority of these compounds crystallize in the highest-symmetrical tetragonal structures [20,25–27].</sub>

The [*LMO*<sub>2</sub>]*X* stoichiometry of the X1 compounds (*L* = Ca, Cd, Sr, Pb, Ba; *M* = Sb, Bi) may actually lead to all three structures shown in Fig. 1. The ordered PbSbO<sub>2</sub>Cl structure seems to be realized when the *L*<sup>2+</sup> cations are noticeably larger than *M*<sup>3+</sup>, and the *X*<sup>-</sup> anion is, on the contrary not much larger than both *L*<sup>2+</sup> and *M*<sup>3+</sup>, i.e., chloride. If 1/2(*R*(*L*<sup>2+</sup>) + *R*(*M*<sup>3+</sup>)) ≪ *R*(*X*<sup>-</sup>), the disordered Nd<sub>2</sub>O<sub>2</sub>Te structure becomes more favorable, which is best demonstrated by the PbSbO<sub>2</sub>*X* and CaBiO<sub>2</sub>*X* series, where both oxyhalides and CaBiO<sub>2</sub>Br adopt the Nd<sub>2</sub>O<sub>2</sub>Te structure.

The other, as yet very rare, version of cation-ordered X1 structure, adopted hitherto only by CaBiO<sub>2</sub>Cl and CdBiO<sub>2</sub>Cl, was assumed to occur due to the small size

Table 3  
Atomic coordinates for the new bismuth oxyhalides

(a) Nd <sub>2</sub> O <sub>2</sub> Te type (CaBiO <sub>2</sub> I)							
Atom	Site	<i>z</i>	<i>U</i> × 100				
Ca <sub>0.5</sub> Bi <sub>0.5</sub>	4 <i>e</i> (0,0, <i>z</i> )	0.3290(2)	2.0(1)				
O	4 <i>d</i> (0,1/2,1/4)	—	2.6(4)				
I	2 <i>a</i> (0,0,0)	—	3.0(1)				
(b) PbSbO <sub>2</sub> Cl type							
Compound		SrBiO <sub>2</sub> Br		SrBiO <sub>2</sub> I		BaBiO <sub>2</sub> I	
Atom	Site	<i>x</i> ( <i>y</i> )	<i>U</i> × 100	<i>x</i> ( <i>y</i> )	<i>U</i> × 100	<i>x</i> ( <i>y</i> )	<i>U</i> × 100
<i>L</i>	4 <i>c</i> (0, <i>y</i> ,1/4)	0.3943(4)	0.9(1)	0.4050(3)	1.8(1)	0.3971(2)	1.8(1)
Bi	4 <i>c</i> (0, <i>y</i> ,1/4)	0.0796(2)	1.8(1)	0.0739(2)	2.3(1)	0.0689(2)	1.6(1)
O	8 <i>e</i> ( <i>x</i> ,0,0)	0.247(2)	2.0(6)	0.247(1)	2.6(4)	0.251(2)	3.2(4)
<i>X</i>	4 <i>c</i> (0, <i>y</i> ,1/4)	0.7456(4)	2.3(2)	0.7445(2)	3.1(1)	0.7480(2)	2.6(1)

Table 4  
Selected bond distances for the new bismuth oxyhalides

Compound	CaBiO <sub>2</sub> I	SrBiO <sub>2</sub> Br	SrBiO <sub>2</sub> I	BaBiO <sub>2</sub> I
<i>L</i> –O	2.265(1) × 4	2.442(9) × 4	2.432(5) × 4	2.543(6) × 4
<i>L</i> – <i>X</i>	3.631(2) × 4	3.345(4) × 2 3.458(4) × 2	3.511(3) × 2 3.652(3) × 2	3.552(3) × 2 3.677(2) × 2
Bi–O	2.265(1) × 4	2.249(9) × 4	2.272(5) × 4	2.315(6) × 4
Bi– <i>X</i>	3.631(2) × 4	3.605(3) × 2 3.584(3) × 2	3.735(2) × 2 3.776(2) × 2	3.881(2) × 2 3.931(3) × 2

of Ca<sup>2+</sup>, which is only seven-coordinated, contrary to eight-fold coordination in both other X1 structures [9], coupled with the smaller size of Cl, leading to only six-coordination of the anion. In contrast, Ca<sup>2+</sup> is eight-coordinated both in CaBiO<sub>2</sub>Br and in CaBiO<sub>2</sub>I involving larger halide ions.<sup>2</sup>

The influence of electronic structure of the *L* cation is an overall factor governing existence of the compound itself and its stoichiometry hence the overall possibility of cation ordering. It is thus more interesting to compare structures of compounds where *L*<sup>II</sup> ions have similar radii but different electronic structures. The ionic radii of Ca<sup>2+</sup> and Cd<sup>2+</sup> being almost equal [25] and neither of these cations having a lone pair, it is not surprising that compounds of these metals are isostructural. Lead compounds behave differently from their strontium analogs, though the difference in radii of Pb<sup>2+</sup> and Sr<sup>2+</sup> is rather small. In the PbBiO<sub>2</sub>*X* series, only the oxychloride may exist in the ordered form, the

<sup>2</sup>Our investigations in the area of isostructural antimony compounds have shown that a new compound CaSbO<sub>2</sub>Cl crystallizes in the PbSbO<sub>2</sub>Cl structure with cell parameters *a* = 5.658(2) Å, *b* = 12.114(4) Å, and *c* = 5.442(1) Å. In this case, Ca<sup>2+</sup> is also probably eight-coordinated, but the cation ordering agrees with that expected based on the Ca<sup>2+</sup>/Sb<sup>3+</sup> size contrast, and chloride is expected to be only four-coordinated.

others are disordered; this most likely occurs due to the same configuration of Pb<sup>2+</sup> and Bi<sup>3+</sup> (6*s*<sup>2</sup>). It would be interesting to compare the PbSbO<sub>2</sub>*X* with their strontium analogs, as yet unknown. A systematic study of related antimony (and possible arsenic) oxyhalides could give a deeper insight into crystal chemistry of X1 type oxyhalides. Some of these investigations are now in progress.

## Acknowledgments

This work was carried out as a part of Russian Foundation of Basic Researches Project No. 01-03-38006a.

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